AI-Ca-Mg (Aluminum-Calcium-Magnesium)

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The early literature on this ternary system, which includes the work of [1958Cat], is summarized by [1995Vil], who presented partial isothermal sections for Mg-rich alloys at 400, 370, and 290 °C. A number of recent publications [1995Nin, 2001Ozt, 2003Ame, 2003Gro, 2003Ozt, 2003Tka, 2004Zho, 2005Isl, 2005Par, 2005Suz, 2005Zho, 2006Suz, 2007Alj] have appeared on this system with focus on Mg-rich alloys.

Binary Systems

The Al-Ca system was experimentally reinvestigated by [2001Kev1] and assessed thermodynamically by [2001Kev2]. There are four intermediate phases in this system: Al₄Ca (*D*1₃, Al₄Ba-type tetragonal), Al₂Ca (*C*15, MgCu₂-type cubic), AlCa (or Al₁₄Ca₁₃; monoclinic, space group *C*2/*m*), and Al₃Ca₈ (Ca₈In₃-type triclinic, space group *P*1). The Al-Mg phase diagram [2003Cze, Massalski2] has the following intermediate phases: Mg₂Al₃ (cubic, denoted β), ϵ or R (rhombohedral), and Mg₁₇Al₁₂ (*A*12, α Mn-type cubic, denoted γ). The Ca-Mg phase diagram [1995Aga] has one intermediate phase Mg₂Ca (*C*14, MgZn₂-type hexagonal). A thermodynamic reassessment of this system was reported recently by [2006Zho].

Ternary Phases

There are several reports on the existence of a C36, MgNi₂-type hexagonal ternary phase along the Al₂Ca-Mg₂Ca join [2003Ame, 2005Suz, 2005Zho, 2006Suz].

[2003Ame] found a homogeneity range from CaAl_{1.34} Mg_{0.66} to CaAl_{0.93}Mg_{1.07} and the corresponding lattice parameter ranges of $a = 0.5835 \cdot 0.5935$ nm and $c = 1.8897 \cdot 1.9258$ nm. However, [2006Suz] reported a stoichiometry close to CaAl_{1.33}Mg_{0.67}. The existence of the C36 phase appears to be established and the absence of this ternary phase in the earlier reports is probably due to the difficulty of distinguishing the x-ray patterns of the C14 and C36 phases. The existence of a second ternary phase Al₂(Mg, Ca) with Mg concentration range of 17.5-22.5 at.% was reported by [2005Zho]. This phase needs confirmation.

The Liquidus Surface

Calculated liquidus projections for the entire composition range of this ternary system presented by [2003Gro] and [2005IsI] are in agreement, but both calculations ignored the existence of the ternary C36 phase. Partial experimental liquidus projections for Mg-rich alloys were presented by [2003Tka] and [2005Suz]. The primary crystallization of the C36 phase was not reported by [2003Tka]. The partial projection from [2005Suz] is shown in Fig. 1. The phases of primary crystallization in the Mg-rich alloys are (Mg), γ , C14, and C36. Among the computed liquidus projections [2001Ozt, 2003Gro, 2003Ozt, 2004Zho, 2005IsI], only the projection by [2004Zho] shows a primary field of the C36 phase.

Figure 2 and 3 show the partial isothermal sections at 500 and 400 °C determined by [2006Suz]. At 500 °C (Fig. 2), the C36 phase forms tie-lines with C14 (Mg₂Ca), C15 (Al₂Ca),



Fig. 1 Al-Ca-Mg partial liquidus projection [2005Suz]



Fig. 2 Al-Ca-Mg partial isothermal section at 500 °C [2006Suz]



Fig. 3 Al-Ca-Mg partial isothermal section at 400 °C [2006Suz]

and (Mg). A three-phase field of (Mg) + C14 + C36 was identified by [2006Suz]. At 400 °C (Fig. 3), the C36 phase is not stable. (Mg) forms tie-lines with C14, C15, and Mg₂Al₃(β) phases. Three-phase fields of (Mg) + C14 + C15 and (Mg) + C15 + β were identified. A narrow three-phase field of ($\epsilon + \beta + \gamma$) is probably present along the Al-Mg side.

Very recently, [2007Alj] carried out an experimental investigation of this system, using 21 ternary alloys covering the entire composition range. The phase equilibria were studied with differential scanning calorimetry, x-ray powder diffraction, and metallography. The results were compared with vertical sections computed from the thermodynamic description of [2005IsI]. The C36 phase was not found in the experiments of [2007Alj] and was not included in the thermodynamic description of [2005IsI]. The comparison of the DSC data with the computed liquidus showed agreement in some cases, but discrepancies were found in other cases.

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